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Study of Structural Modification in $\text{YBa}_2\text{Cu}_3\text{O}_{7-y} + x \text{Sb}_2\text{O}_3$ Superconductor Composite

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Abstract

A series of $\text{YBa}_2\text{Cu}_3\text{O}_{7-y} + x \text{Sb}_2\text{O}_3$ ($x = 0.0, 0.01, 0.03, 0.05, 0.08, 0.2$ wt. %) samples were prepared using the solid state reaction method. The structural disorder has been studied using XRD and Raman techniques. XRD graphs show the unchanged orthorhombic structure. The increase in peak width of the XRD pattern in the composites signifies the decrease in crystallite size with Sb_2O_3 addition. Raman analysis confirms all the vibration modes in pure YBCO along with some peaks of antimony. The structural changes are marked by peak sifting in the XRD pattern and the softening in the Raman modes. Linear relation is portrayed between oxygen content and the c axis lattice parameter. The c axis tends to increase with the decrease in oxygen content as calculated from XRD and Raman data. EDX analysis shows the incorporation of Sb_2O_3 to grain boundary region along with the presence of all the compositional elements. Microstructure analysis has been studied through SEM. SEM graphs shows that pristine YBCO exhibits large and elongated grains randomly oriented in all directions. The grain size decrease in low wt. % addition of Sb_2O_3 and increase in higher wt. % addition.

Keywords: HTSC, RAMAN, Electrical conductivity, Grain boundary, Microstructure.

Introduction

Even after several decades of research on high T_c superconductors (HTSC), these materials are still interesting for many researchers. In recent years there have been intense theoretical and experimental activities in the investigation of the copper oxide based high T_c superconductors¹⁻³. Bulk HTSC samples are granular in nature, characterized by the presence of grains and twin

boundaries, secondary phases and other less important defects. Thus, granular superconductors can be considered as disordered systems formed by anisotropic grains weakly coupled and randomly distributed, where we can distinguish two main contributions to their properties. The first one, called intra-grain contribution, is associated with high critical current density values and grains completely shielded at sufficiently low magnetic fields. The second one, called inter-grain contribution, is characterized by lower critical current density values associated to Josephson couplings. Thus, the granularity arising from this structure, is responsible for the typical low J_c values of HTSC and for their strong dependence with temperature and applied magnetic field. Therefore, the inter-grain material has an important role in the technological applications of HTSC, in particular in the presence of

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magnetic fields⁴⁻⁶. The planes containing copper and oxygen atoms that are chemically bonded to each other have a crucial importance in YBCO. The special nature of the copper-oxygen chemical bond contributes to an effective electrical conductivity in certain directions. The existence of the Cu-O chains and the CuO₂ planes in cuprate superconductors indicates the important role of Cu atoms. Thus, the ionic radius as well as the valence number of the dopant element has a crucial role in determining the characteristics of new superconductors. Doping of YBCO with various elements is conducted for two basic purposes. The first one is the modification of the microstructure in order to obtain fundamental information related to the possible mechanisms, and the second one is to improve its physical characteristics. It has been seen that substitutions at the Y and Ba sites does not generate any discernible effect, while substitution at the Cu and O sites has significant effects on the superconducting properties of cuprates⁷. Due to the low melting point and the possible surfactant action of semi-metallic Sb₂O₃ doping, it is expected that doping with Sb could improve the grain boundary characteristics and enhance the inter-granular coupling in Y123. Paulose et al.⁸ reported that Sb₂O₃ doping significantly increases the rate of oxygen absorption in the Y123 system. Jin et al.⁹ determined that Sb₂O₃ doping of Y123 improves the value of the critical current density, J_c, along with a decrease in T_c and a reduction in grain size with increasing porosity. Also Vlahov¹⁰ reported that even small doping levels of Sb shift the step of J_c (B) dependence to higher magnetic fields but cannot improve the weak link behaviour.

Experimental Procedure: YBCO powder is prepared by the solid state reaction route by mixing stoichiometry amount of Y₂O₃, BaCO₃, CuO followed by grinding, calcination at 900⁰ C, sintering at 920⁰C and annealing at 500⁰C for 8 hrs for oxygen uptake respectively. A series of polycrystalline composite samples of (1-x) YBCO + x Sb₂O₃ (x = 0.0, 0.01, 0.03, 0.05, 0.08, 0.2 wt. %) were ground and pressed into pellets. The composite pellets were sintered at 920⁰ C for 12 hours and then cooled to 500⁰ C where they were kept for 5 hours in an oxygen atmosphere for oxygen intake. Phase confirmation and structural analysis were done by X-ray powder diffraction technique (Rigaku Ultima IV set-up). Raman study was done to confirm the vibrational modes (Renisaw invia, Laser (514nm & 785nm) with Leica microscopy DM-2500M). The grain morphology of the samples was

analyzed by scanning electron microscope (Model No. JSM-6480 LV, Make JEOL) and the compositional analysis was determined by energy dispersive X-ray analysis (EDX) using an INCA Oxford Analyzer attached to a scanning electron microscope.

Results and Discussion

XRD Analysis: Figure 1 shows the X-ray diffraction patterns of the YBa₂Cu₃O_{7-y}/xSb₂O₃ composite samples. The diffraction pattern of all the samples is indexed using ChekCell software and the results are found to be in orthorhombic phase with a space group P_{mmm}, with some Sb₂O₃ peaks. The lattice parameters and unit cell volumes of these samples are presented in Table1.

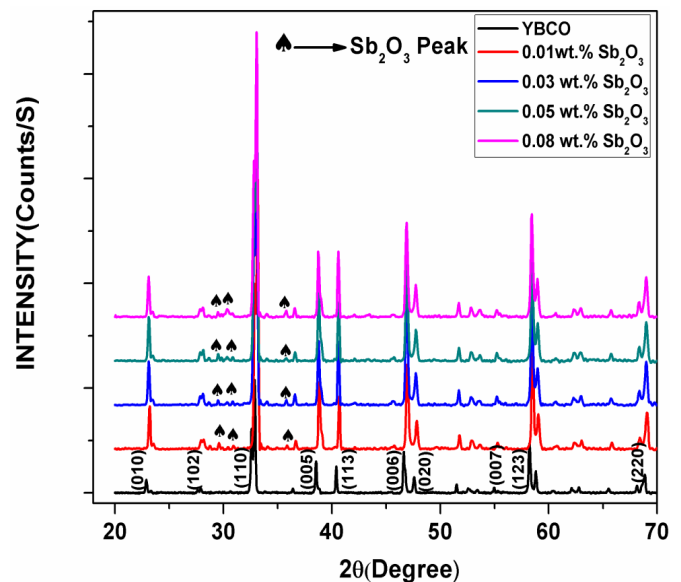


Figure 1. XRD Patterns of YBCO+ x Sb₂O₃ Samples (x = 0.0, 0.01, 0.03, 0.05, 0.08 wt. %).

Figure 2 shows magnified (005) and (006) planes. The peak intensity increases monotonically for Sb₂O₃ added samples. The clear shifting of the peak position to the higher angle side is visible in all the composite samples. Basically the crystallite size is reflected in the broadening of a particular peak in a diffraction pattern associated with a particular planar reflection from within the crystal unit cell. It is inversely related to the FWHM of an individual peak. The narrower the peak, the larger the crystallite size. The FWHM value is smaller in the pristine sample as compared to the composite samples. The peak width little bit increases in the composites which signifies the decrease in crystallite size with Sb₂O₃ addition.

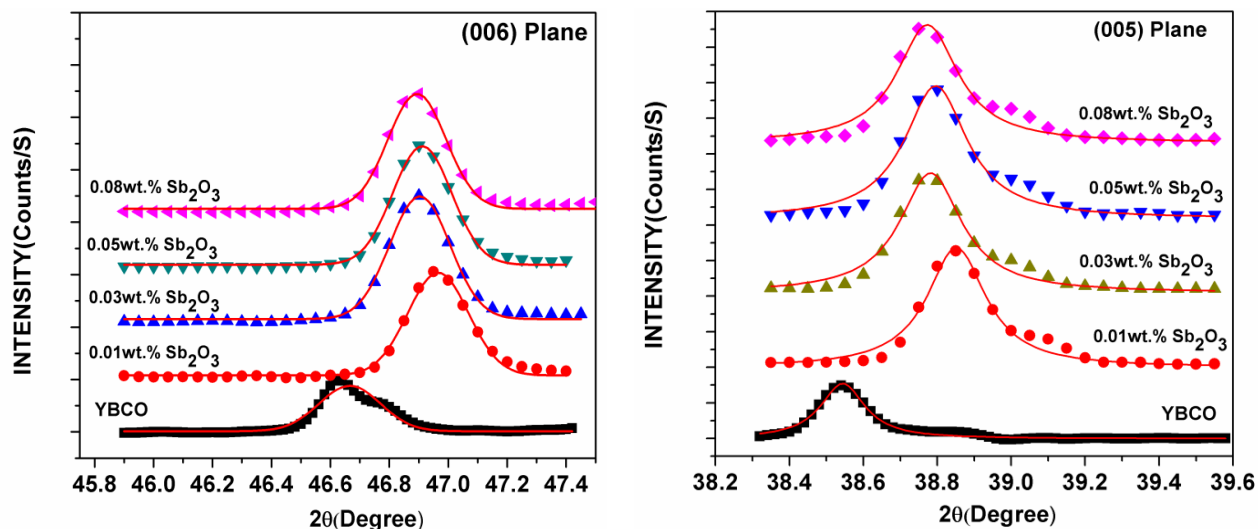


Figure 2. Lorentzian fitting of (005) and (006) plane plotted by varying Sb_2O_3 wt.%.

Table 1. Parameters calculated from XRD graphs

Sb_2O_3 (wt.%)	a (Å)	b (Å)	c (Å)	V(Å^3)	Anisotropy
YBCO	3.818(1)	3.884(1)	11.681(2)	173.218	1.7
0.01	3.824(3)	3.885(2)	11.703(5)	173.862	1.5
0.03	3.818(2)	3.874(1)	11.728(7)	173.468	1.4
0.05	3.804(2)	3.884(1)	11.686(6)	172.657	2.0
0.08	3.813(1)	3.871(1)	11.658(4)	172.073	1.5

The length of 'b' parameter remains almost same in the composite samples except the 0.03 and 0.05 wt.% Sb_2O_3 added sample, where the length decreases slightly. The length of 'c' parameter is higher in the composite samples as compared to the pristine sample except the highest wt.% added sample where the length decreases slightly. For low oxygen content sample, the c-axis length is high. The oxygen stoichiometry value is calculated from the relation $7-y = 75.250 - 5.856c$ (presented in Table 2) where c is the c-axis lattice parameter¹¹. The anisotropy, $\text{Anis} = 100 (b-a)/0.5 (b+a)$ which is the percentage deviation from tetragonal structure is calculated and listed in Table 1. The anisotropy is 1.7 for pristine and highest for 0.05 % Sb_2O_3 added sample.

RAMAN analysis: Raman spectroscopy is a characterization method that measures the frequencies of the long-wavelength lattice vibrations (phonons). Figure 3 shows the Raman spectrum of $\text{YBCO} + x \text{Sb}_2\text{O}_3$ ($x = 0.0, 0.01, 0.03, 0.05, 0.08$ wt. %) composite samples. A

perovskite YBCO unit cell has five different oxygen sites which get activated under the influence of photons: O (1) in CuO chains, O(2,3) in CuO_2 planes, O(4) in the apical chains and O(5) lies in between the oxygen chains which is empty in orthorhombic structure. The active vibration along the c axis are shown at peak position 500 cm^{-1} , 432 cm^{-1} , 339 cm^{-1} . 500 cm^{-1} represents the stretching of apical oxygen or the bridging oxygen O(4) denoted as $\text{O}(4)\text{A}_g$. 432 cm^{-1} represents in phase vibration of O (2)–O(3) oxygen atom in CuO_2 plane marked as $\text{O}(2,3)\text{A}_g$ and 339 cm^{-1} represents out-of-phase c axis vibration of O(2)–O(3) oxygen atom in CuO_2 plane labeled as $\text{O}(2,3)\text{B}_{1g}$ ¹²⁻¹⁵. The other two Raman active modes are vertical along the c axis given by Cu (2) atoms (151 cm^{-1}) denoted as $\text{Cu}(2)\text{A}_g$ i.e plane Cu in phase and Ba atoms (115 cm^{-1}) marked BaA_g i.e Ba vibration. The strong peak of 500 cm^{-1} at $y=0$ shifts to the lower frequency side as the oxygen content decreases. In addition, the 115 cm^{-1} peak of the $y = 0$ sample disappears at $7-y = 6.2$ ¹⁶.

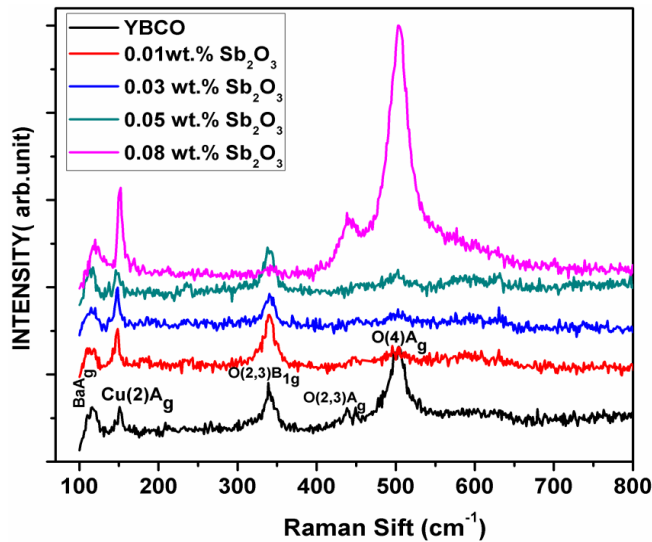


Figure 3. Micro-Raman spectra of YBCO + x Sb₂O₃ Samples (x = 0.0, 0.01, 0.03, 0.05, 0.08 wt.%) samples.

From the Raman graphs (fig3) it is clear that all the vibrational modes are present in the pristine YBCO as well as in the composite samples. The 500 cm⁻¹ peak softens in

0.01, 0.03, 0.05 wt.% added samples and hardens for the highest Sb₂O₃ %. Also the 115 cm⁻¹ and 151 cm⁻¹ peaks hardens in the highest wt.% composite sample. There are some extra modes detected in the doped samples. These are at 188cm⁻¹, 595cm⁻¹, 243cm⁻¹ for 0.01 %, 596 cm⁻¹ for 0.03 %, 241 cm⁻¹, 595cm⁻¹ for 0.05 % and at 213 cm⁻¹ for 0.08 % Sb₂O₃ added samples. These modes are found to be of Sb₂O₃. Generally, the two common forms of Sb₂O₃ are the cubic phase senarmontite and the orthorhombic phase valentinite. It has signals at 82, 118, 141, 189, 219, 254, 295, 355, 373, 450, 500, 596, 684, 712, and 904 cm⁻¹. All signals below 400 cm⁻¹ belong to the external lattice mode regime, while those above 400 cm⁻¹ belong to the internal vibrations^{17,18}. Generally, the O4 phonon mode, is associated with the oxygen content y. Huong et al. proposed the following relationship between the oxygen content and the peak frequency of the O4 mode as,¹⁹

$$y = 13.58 - 0.027v$$

where v is the peak frequency of the O4 mode.

Table 2: Different parameters calculated using XRD and Raman spectra. ‘c’ is lattice parameter along Z direction, y is oxygen loss, v is frequency of apical oxygen.

Sb ₂ O ₃ (wt.%)	c (Å)	O _{7-y} (from XRD)	v(cm ⁻¹)	O _{7-y} (from Raman)
0.00	11.681(2)	6.85	504	7.02
0.01	11.703(5)	6.71	498	6.86
0.03	11.728(7)	6.57	502	6.97
0.05	11.686(6)	6.82	502	6.97
0.08	11.658(4)	6.98	504	7.02

Linear relation is portrayed between oxygen content and the c axis lattice parameter. The c axis tends to increase with the decrease in oxygen content. Table 2 shows the oxygen content (y) evaluated using frequency of Raman mode and the c axis of XRD. It gives an insight of oxygen suppression occurring on the apical site and lengthening of the c axis due to Sb₂O₃ addition. Another thing was noticed here is that, even if the oxygen value remains high in all the pure as well as composite samples, the T_c value reduces to a large extent in the Sb₂O₃ added samples. This signifies that the oxygen concentration is

not the predominant factor for high T_c superconductivity in YBCO system. Its role is to adjust the coupling strength between positive and negative ions.

Microstructural analysis: The grain size distribution of the pristine and doped samples are shown in figure 4. It shows that pristine YBCO exhibits large and elongated grains randomly oriented in all directions. For 0.01 and 0.03 wt.% addition of Sb₂O₃ the grain size decreases to a greater extent. But further increase of Sb₂O₃ addition, the grain size increases and also the porosity increases.

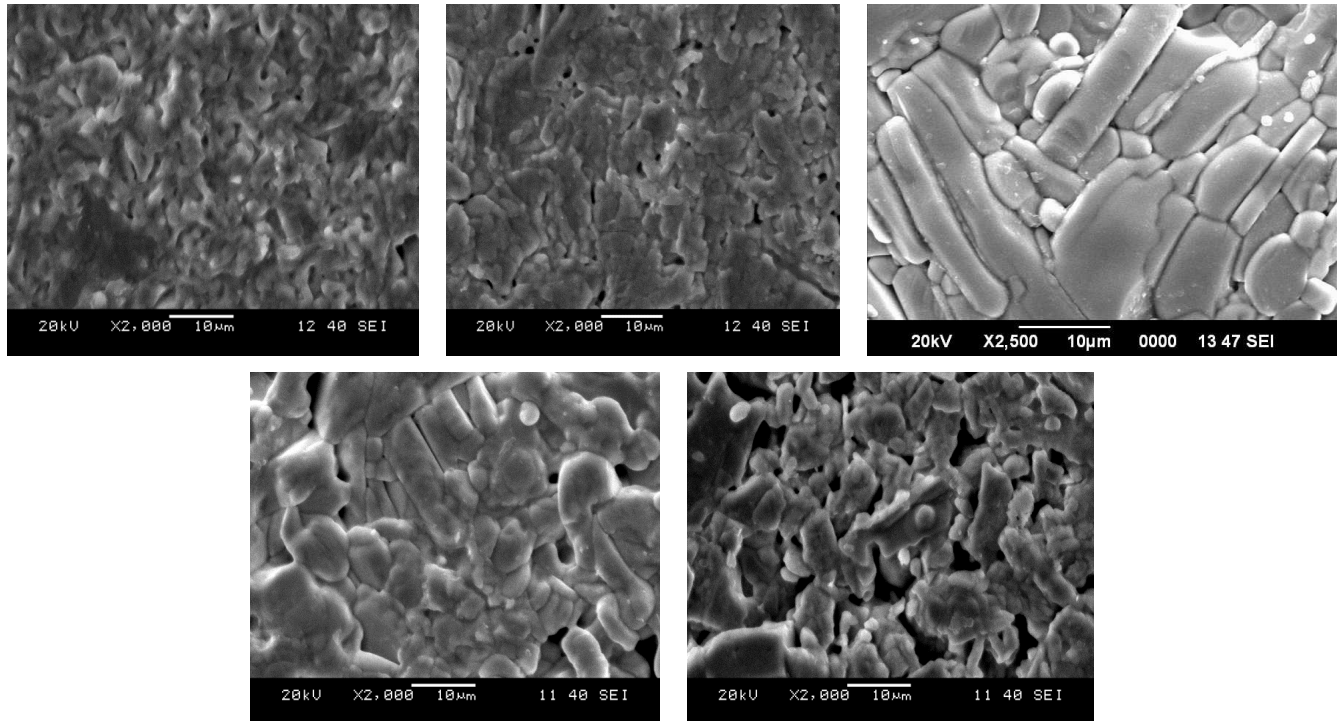


Figure. 4 SEM micrographs of YBCO + x Sb₂O₃ (x = 0.0, 0.01,0.03,0.05,0.08 wt.%) marked as a, b, c, d, e respectively.

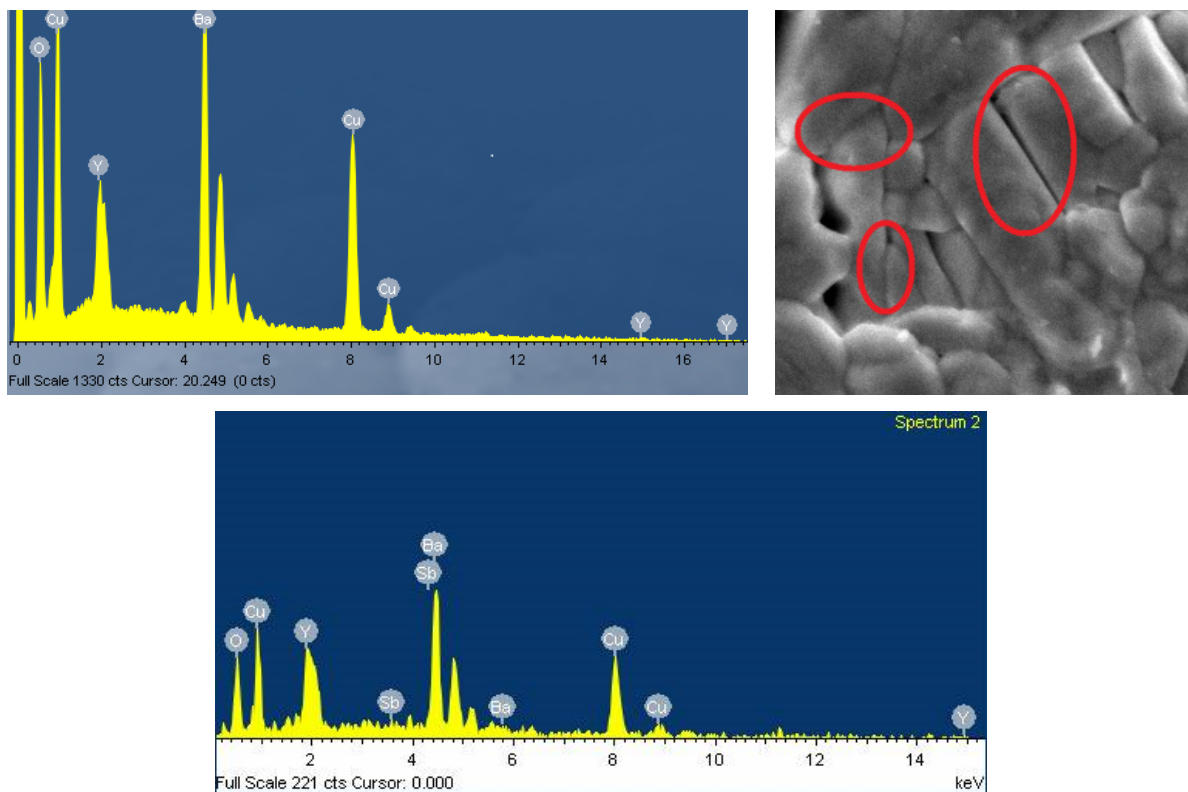


Figure 5. EDX graphs of YBCO and 0.05 % Sb₂O₃ added sample (taken at the grain boundary region) marked as a and b respectively.

The presence of all the compositional elements in the pristine as well as in doped samples is confirmed from the EDX graphs. The EDX taken at the grain boundaries clearly shows the presence of Sb at the boundaries (shown by red circles in the inset SEM fig. 5b). It indicates that some amount of the added antimony goes to the grain boundary region in addition of its incorporation to inside grains, which enhances the inter-grain coupling which is also observed in the R-T plots.

Conclusion

The effect of semiconductor Sb_2O_3 on the structural property is studied. XRD graphs show the unchanged orthorhombic structure in the composite samples. Raman studies show all the vibrational modes in the pristine as well as composite samples in addition to presence of Sb_2O_3 modes. The structural changes are marked by peak shifting in the XRD pattern and the softening in the Raman modes. SEM micrographs show reduced grain size in low % addition of Sb_2O_3 and grain size increases for the high % addition of Sb_2O_3 . EDX analysis shows the incorporation of Sb_2O_3 to grain boundary region along with the presence of all the compositional elements.

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Ethical Clearance: NA

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